Energetics of hydration on uranium oxide and peroxide surfaces

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Enthalpies of water adsorption on amorphous and crystalline oxides and peroxides of uranium are reported. Despite substantial structural and computational research on reactions between actinides and water, understanding their surface interactions from the energetic perspective remains incomplete. Direct calorimetric measurements of hydration energetics of nano-sized, bulk-sized UO\textsubscript{2}, U\textsubscript{3}O\textsubscript{8}, anhydrous \(\gamma\)-UO\textsubscript{3}, amorphous UO\textsubscript{3}, and U\textsubscript{2}O\textsubscript{7} were carried out, and their integral adsorption enthalpies were determined to be \(-67.0\), \(-70.2\), \(-73.0\), \(-84.1\), \(-61.6\), and \(-83.6\) kJ/mol water, with corresponding water coverages of 4.6, 4.5, 4.1, 5.2, 4.4, and 4.1 H\textsubscript{2}O per nm\textsuperscript{2}, respectively. These energetic constraints are important for understanding the interfacial phenomena between water and U-containing phases. Additionally, this set of data also helps predict the absorption and desorption behavior of water from nuclear waste forms or used nuclear fuels under repository conditions. There are also underlying relations for water coverage among different U compounds. These experimentally determined data can be used as benchmark values for future computational investigations.

Introduction

The interaction of uranium phases with water is an important topic as it is closely linked to nuclear fuel storage, repository research, and environmental concerns of nuclear waste contamination. Water can interact with actinides in multiple ways, including sorption, radiolysis, and hydrothermal chemical reactions, which result in numerous interesting and important phenomena.

Specifically, water may participate as a catalyst in the oxidation of U(IV) [and Pu(IV)] binary oxides where molecular oxygen alone hardly oxidizes the dioxides \([1, 2, 3, 4, 5, 6, 7, 8, 9, 10]\). Water also enhances the corrosion rate of actinides and associated fission products \([5, 11, 12, 13, 14, 15, 16]\). For instance, rates of corrosion increase greatly when metallic plutonium is exposed to moisture instead of dry air \([11, 17]\).

In the nuclear energy industry, water plays influential roles in the corrosion and degradation of nuclear fuel \([15, 18, 19, 20]\), since H\textsubscript{2}O provides energetically favorable pathways toward more soluble species, such as metastudtite, uranyl peroxide clusters, or other uranyl ions \([15, 16, 21, 22, 23, 24]\). Within used nuclear fuel (UNF) repositories, water also participates in multiple processes including dissolution, hydrogen production, and phase alteration \([6, 25, 26, 27, 28, 29]\). The fate of uranium in a nuclear plant accident is also subject to its interaction with water. For instance, during and in the aftermath of the Fukushima Daiichi nuclear disaster, UO\textsubscript{2} can react with high temperature steam and alkaline sea water to form complex solid phases and solutions \([23, 30, 31]\). One of the current
research efforts in nuclear industry is to develop accident-tolerant fuels such as U-Si phases [32]. Furthermore, water contributes significantly to the formation of many uranyl minerals as weathering products [33]. The chemical and physical sorption of water onto and into minerals and actinide-mineral surface reactivity in aqueous solution are also important [34, 35]. In addition, in the hydrothermal geological settings, water facilitates the alteration of uranium minerals [20, 36, 37], such as the generation of uranyl oxyhydroxides and peroxides in natural U deposits [18, 20, 21, 38, 39].

Because of the daunting issues and needs stated above, investigations of the influence of water on actinide oxides and their potential reactions have increased rapidly during the past decades from both experimental and computational perspectives. Computational techniques can provide insights about the structures and energetics of actinide dioxide-water interfaces [14, 40, 41, 42, 43, 44, 45, 46, 47], which were also studied by advanced structural characterization techniques in recent years [7, 8, 9, 10, 48, 49, 50, 51, 52]. However, related experimental thermodynamic investigations [43, 52, 53] are still scant. A solid set of calorimetric data about surface interactions is therefore of great importance in (i) serving as experimentally determined benchmarks for future computational studies and (ii) illustrating how water starts interacting with UNF and U-based nuclear wastes under a variety of geological environments.

Thus, in this work, we performed direct calorimetric measurements of water adsorption on the surfaces of UO2, anhydrous γ-UO3, amorphous UO3, and amorphous U2O7 at room temperature, using a microcalorimeter coupled with a gas adsorption analyzer. Among these samples, we placed a special emphasis on UO2 due to the significant relevance of its hydration energetics in the storage of UO2-based fuels and raw materials, and the permanent disposal of UNF. Previously, its isostructural (fluorite-structured) oxides, CeO2, Cet...U2O2+δ and ThO2 have been studied computationally [43, 52], and HfO2, ZrO2, CeO2, Cet...U2O2+δ, and UO2 have been investigated using calorimetry [52, 54, 55]. Uranyl-containing oxides, unlike UO2, containing highly soluble uranyl ions and are much more susceptible to the release of radionuclides into the environment. Last, amorphous uranyl phases, mostly originating from the thermal deposition of studdite or metastudtite, can also provide pathways for mobilizing uranium in the form of uranyl species from the corroded surface of UNF [15, 16, 21, 56]. For instance, metastudtite and amorphous U2O7 have been discovered as products of the alteration of UNF or from precipitation during mining and storage [16, 18, 22, 56].

The samples in the present study cover almost all important basic uranium-containing phases encountered in the nuclear industry. We report hydration energetics measured by water adsorption calorimetry to address the following three questions: (i) what are the enthalpies of water—uranium oxides interfacial interactions? (ii) what are the differences among the phases? and (iii) how do intrinsic structural variations and/or surface chemistry govern such differences?

Results and discussion

Table I summarizes the measured data. The surface area of anhydrous γ-UO3, 1.7 m2/g, is close to those of bulk UO2 and U3O8, 1.5 and 3.0 m2/g, respectively. Nano-sized UO2 has a much larger surface area (57.9 m2/g). The amorphous compounds (am-UO3, am-U2O7) have similar surface areas, 8.5, and 8.8 m2/g, respectively. We noticed that these decomposed phases have larger surface areas by 7–11% compared to that of metastudtite, and almost quadruple those values of other bulk-sized uranium oxides. All three U(VI)-containing oxides show large values of water adsorption enthalpies for the first dose at near zero coverage: −156.1, −125.1, and −202.9 kJ/mol water, for γ-UO3, am-UO3, and U2O7, respectively. Generally, the enthalpy of adsorption becomes less exothermic as water coverage increases (see Figs. 1–3). In the adsorption enthalpy curves, as the water coverage increases,

<table>
<thead>
<tr>
<th>Sample</th>
<th>Surface area (m2/g)</th>
<th>Coverage (H2O per nm2)</th>
<th>ΔH_ads (kJ/mol)</th>
</tr>
</thead>
<tbody>
<tr>
<td>am-UO2</td>
<td>8.8 ± 0.1</td>
<td>4.1</td>
<td>−83.6</td>
</tr>
<tr>
<td>am-UO3</td>
<td>8.5 ± 0.6</td>
<td>4.4 ± 0.5</td>
<td>−61.6 ± 6.7</td>
</tr>
<tr>
<td>γ-UO3</td>
<td>1.7 ± 0.1</td>
<td>5.2 ± 0.6</td>
<td>−84.1 ± 1.1</td>
</tr>
<tr>
<td>U2O7</td>
<td>3.0 ± 0.1</td>
<td>4.1 ± 0.4</td>
<td>−73.0 ± 1.4</td>
</tr>
<tr>
<td>Bulk-UO2</td>
<td>1.5 ± 0.1</td>
<td>4.5 ± 0.1</td>
<td>−70.2 ± 1.2</td>
</tr>
<tr>
<td>Nano-UO2</td>
<td>5.79 ± 0.2</td>
<td>4.6</td>
<td>−67.0</td>
</tr>
</tbody>
</table>

Figure 1: Differential enthalpies of water adsorption (ΔH_ads) as a function of water coverage on U6+ containing compounds.
differential enthalpies of water adsorption ($\Delta h_{\text{ads}}$) at $-44$ kJ/mol water define the end of the investigation of adsorbed water on these samples. This is because $-44$ kJ/mol water is the heat released when water vapor condenses to the liquid phase at 25 °C. If the hydration energy appears less exothermic than $-44$ kJ/mol water, it is very likely that additional dosed water molecules barely interact with the sample surface, rather tending to self-interact by forming clusters [57, 58]. Such behavior is not relevant to the present study and will not be further discussed. Thus, we can derive the integral water adsorption enthalpy by integrating all $\Delta h_{\text{ads}}$ smaller than $-44$ kJ/mol and the obtained results are shown in Table I.

Starting from UO$_2$, increasing the valence of U will be accommodated by excess oxygen existing as clusters in the interstitial sites of the fluorite structure till the formation of intermediate phases, such as U$_3$O$_7$ and U$_4$O$_9$ [59, 60]. Exceeding O/U = 2.33 will break the fluorite-derived arrangement of cations and lead to the emergence of U$_3$O$_8$ phase [60, 61]. Such oxidizing transitions are the combination of diffusion-controlled processes and nucleation reactions, which has impacts on the hydration energetics of the forming phases. Moving from tetravalent U to mixed valence states, then to hexavalent U, the hydration profiles of the crystalline oxide phases (UO$_2$, U$_3$O$_8$, and UO$_3$) shift accordingly (Fig. 1), with an increasingly exothermic trend. Compared with U$^{4+}$ in UO$_2$, U$^{5+}$ and/or U$^{6+}$ in U$_3$O$_8$ and UO$_3$ provide increased electronic attractions to the negatively charged oxygen in water molecules, leading to more stable U-O bonding [60, 62]. Particularly, the integral enthalpy of water adsorption of UO$_2$ is $-70.2 \pm 1.2$ kJ/mol water, relatively close to that of CeO$_2$ ($-59.8 \pm 0.7$ kJ/mol water) or ThO$_2$ ($-65.0 \pm 1.2$ kJ/mol water) [43], both of which also crystalize in the fluorite structure, but Ce or Th cannot be further oxidized.

The oxidation of UO$_2$ in humid conditions may result in oxidation of the near surface (up to 1 μm) and/or formation of surface hydroxyls, observed by XPS and Raman spectroscopy [7, 8, 48, 49]. Water vapor alone has only minimal (catalytic) impacts on the surface oxidation of UO$_2$ [10, 14, 59]. In Fig. 2, there is a relatively sharp transition of the differential enthalpy of adsorption curve of UO$_2$ from overlapping that of U$_3$O$_8$ (step i) to that of γ-UO$_3$ (step ii). U$_3$O$_8$ is the weathering product of uraninite and it is kinetically and thermodynamically stable. Previously, it has shown that the oxidation and hydration of U$_3$O$_8$ to schoepite can occur only after long storage time (2–3.5 years) in humidity [63], suggestive of a very slow reaction rate. Therefore, U$_3$O$_8$ samples under current experimental periods should not have any noticeable bulk-scale structural alterations. Thus, we conclude before the coverage of 3 water per nm$^2$, water adsorption on UO$_2$ behaves similarly to that on the reduced U oxide. Suggested by DFT calculations and experimental observations, the (111) surface of UO$_2$ has the lowest surface energy [43, 44, 64, 65] that has the highest possibility to stabilize the molecular or dissociated water molecules. Thus, the interaction enthalpy from the initial water dosage $\Delta h_{\text{ads}}$ = $-103.4 \pm 7.7$ kJ/mol can be attributed to the interaction between the (111) surface and water consistent with calculated results ($-90.7 \sim -106.1$ kJ/mol) [43, 44, 45, 46]. The difference may be due to the energetic contributions from other planes or defective adsorption sites. U$_3$O$_8$ can be viewed as the high temperature phase resulting from the shear transformation of the UO$_2$ lattice along the (111) planes [61, 66]. Thus, UO$_2$ and U$_3$O$_8$ have similar atomic arrangements in (111) layers which could lead to the similarity in adsorption behavior and integral water adsorption enthalpies ($-70.2$ versus $-73.0$ kJ/mol). However, as the water coverage increases into the “step 2” regime, additional water molecules may interact with the altered surface modified by previous adsorbed water that comprised higher valence U and hydroxyl groups, which could resemble the surface of γ-UO$_3$ under the same water coverage (Fig. 2). Therefore, the measured water adsorption enthalpy of UO$_2$, $\Delta h_{\text{ads}} = -70.2 \pm 1.2$ kJ/mol, includes the energetic terms from the oxidation of U$^{4+}$ and the interaction of water with modified surfaces. This explains why $\Delta h_{\text{ads}}$(UO$_2$) is higher than that of CeO$_2$ ($-59.8 \pm 0.7$ kJ/mol) [55] or ThO$_2$ ($-65.0 \pm 1.2$ kJ/mol) [43], where Ce$^{4+}$ and Th$^{4+}$ cannot be further oxidized. The $\Delta h_{\text{ads}}$ of UO$_2$ from this work is also consistent with a recently published $\Delta h_{\text{ads}}$ of UO$_2$ by Shelly et al. [68 ± 7 kJ/mol] [52]. The observation of stepwise behavior is consistent with previous reports on the humidity-dependent influence of water vapor on the surface of UO$_2$ where three processes were identified in the alteration of fresh UO$_2$ fuel under aerated water and high-temperature conditions [67].

Figure 2: Comparison of differential enthalpies of water adsorption and water coverages among γ-UO$_3$, U$_3$O$_8$, and UO$_2$. 

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For nano-sized UO₂, the enthalpy of water adsorption for the first dose is almost 30 kJ/mol more exothermic than for bulk UO₂ (−137.1 kJ/mol), presumably due to the much higher surface area with more distributed defects and active sites available for bonding. Yet there is a convergence of their adsorption curves at −44 kJ/mol that yields similar water coverages before condensation (Fig. 3): 4.5 and 4.6 H₂O per nm² for bulk-sized and nano-sized UO₂, respectively. This could suggest that despite having a higher surface area, nano-UO₂ still has a hydrophobic surface; and after saturating all active sites by water molecules, the continuing adsorption behavior after around 5 H₂O per nm² coverage will be similar between nano- and bulk-UO₂. Therefore, the hydration process of nano-UO₂ may not differ too much from its bulk counterpart under a high humidity environment.

Conclusions

In this work, we performed direct calorimetric measurements of hydration energetics of nano-sized, bulk-sized UO₂, U₃O₈, anhydrous γ-UO₃, amorphous UO₃, and U₂O₇, and obtained their water adsorption enthalpies to be −67.0, −70.2, −73.0, −84.1, −61.6, and −83.6 kJ/mol water, with water coverage of 4.6, 4.5, 4.1, 5.2, 4.4, and 4.1 H₂O per nm², respectively. Particularly, the hydration behavior of UO₂ was deconvoluted to a two-step process with the first part overlapping that of U₃O₈ then transiting to the second part resembling that of γ-UO₃. The difference between hydration energetics of nano- and bulk-UO₂ resides mostly in low water coverages and diminishes at high water coverages where the two water adsorption energetic profiles converge, which indicates the hydration of UO₂ in the environment over the long term is less dependent of the particle size.

Methods and approaches

The calorimetric experiments were accomplished by a gas adsorption calorimetric method using a Calvet-type microcalorimeter unit (Setaram SenSys) coupled with a commercial gas adsorption analyzer with a gas dosing system (Micromeritics ASAP 2020, Norcross, Georgia) [58, 68] (see Fig. 4). The experiments were performed in four steps. First, a ~80 mg sample was placed in a silica-glass forked tube and degassed under vacuum (<10⁻³ Pa) for 24 h using the Micromeritics ASAP 2020 instrument. Second, the free space of the sample tube and the surface of the sample were determined by Brunauer–Emmett–Teller (BET) analysis of nitrogen, argon or krypton adsorption. Third, the forked tube was inserted into the calorimeter at 25 °C, and water adsorption experiments were programmed with the Micromeritics instrument using 1 μmol of water vapor dose amounts and 1.5 h equilibration time after each dose. The actual adsorbed amounts were obtained from the measured adsorption isotherm. Each dose produced an exothermic peak recorded by the calorimeter software. The integration of the peak provided the enthalpy of adsorption for each dose using a calibration factor based on the enthalpy of fusion of gallium.

The degas temperature prior to the adsorption experiments was selected to retain the structure of studied samples. The UO₂ sample, confirmed to be stoichiometric, was provided by Los Alamos National Laboratory. γ-UO₃ and U₃O₈ were obtained from International Bio-Analytical Industries.
UO$_2$ and U$_2$O$_3$ were obtained by the thermal decomposition of metastudtite [16, 56]. Preheating and degassing treatments were performed in the forked tube using the Micromeritics instrument. UO$_2$ and U$_3$O$_8$ samples were degassed at 400 °C for 4 h. Anhydrous UO$_3$ was formed by heating hydrated UO$_3$ and degassing at 350 °C for 5 h. am-U$_2$O$_2$ and am-UO$_3$ were pretreated at 80 °C for 5 h in vacuum.

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